



when compared with either the main chain length or the PEO used in our swelling measurements.

The NVP used was a fraction which boiled between 57- 58° C at 2 mm. Monomer (I) was prepared by dissolving 2 moles of allyl alcohol together with 1 mol of 2,4 TDI in 290 g CH<sub>2</sub>Cl<sub>2</sub> and refluxing the mixture for three hours. The product is isolated by evaporation of the solvent, redissolved and recovered twice (CH=CH<sub>2</sub> calc. 18.61, found 18.00 %).

Networks were prepared by codissolving NVP, (I) and 0.8 % (based on monomer) of AIBN along with 10 % N- Methylpyrrolidone. These solutions were then injected into closed molds which consisted of two glossy mylar sheets separated by a silicone rubber gasket. The mylar sheets were backed by flat glass plates and the entire assembly was clamped together by strong spring binders. Polymerization at 60° C for two hours yielded polymer sheets (~1 mm thick), which were demolded by immersing the sandwiches in Dry Ice/ methanol. The presence of N- Methylpyrrolidone aided this demolding process. The hydrogels were washed in water for 7 days, the water being changed twice daily. Three hydrogels were prepared in which the starting monomer compositions were 2.00 mole % monomer (I) (gel 1); 1.00 mole % (I) (gel 2); 0.50 mole % (I) (gel 3), respectively. The equilibrium swelling compositions for these materials in water at 25° C are:

$M_n$  of PEO (Union Carbide Corp.) were determined by freezing point depression in aqueous solution and by OH- group titration when appropriate.  $\rho$  was given by the supplier.

Gel	Wt % Gel	$\phi_1$	$\phi_3$
1	18.62	0.8410	0.1590
2	14.82	0.8743	0.1257
3	10.55	0.9112	0.0888

Equilibrium swelling measurements were performed at 25° C in aqueous solutions of PEO. The swollen gel is cut into conveniently sized rectangular pieces of 100-300 mg, blotted dry with tissue paper and weighed to an accuracy of 0.1 mg. The gel pieces are then placed into the PEO solutions and the vials are allowed to gently rotate at 25° C. Typically, the swelling equilibrium was established within 48 h for the lowest  $M_n$  PEO, within 72 h for the intermediate  $M_n$  PEO and within 96 h for the higher  $M_n$  PEO solutions. The hydrogels are then rinsed in fresh water, blotted dry and reweighed to determine the new equilibrium weight. The specimens are then dried at 60° C, under vacuum for 24 hours and weighed in the dry state.

PEO	$\rho$ (g/ml)	$M_n$	$\bar{V}_{12}$
300	1.12	300	0.0675
600	1.12	615	0.0329
750	1.09	750	0.0262
1000	1.10	974	0.0203
2000	1.10	1903	0.0104

### Results and Discussion

Since the hydrogels were placed into the PEO solutions in the swollen state the weight of dry gel is calculated from the known initial swollen weight and the amount of linear PEO absorbed is calculated from the difference between this value and the final dry weight. The control data presented in the Table were obtained by averaging the volume fractions of water and gel over five determinations for each gel in which the standard deviation was less than 0.5 %. The volume fractions were calculated for solution and gel phases at equilibrium taking into account all corrections including loss of linear polymer from solution to gel and exchange of solvent between solution and gel. Since the initial solution volume was more than ten times the initial gel volume the corrections are generally only in the order of 2- 5 %, but were, nevertheless, included in

Table A: Equilibrium conditions for the system water (1) - PEO (linear) (2) - PVP (crosslinked) (3)

PEO	300			600			750			1000			2000		
	Sol V <sub>2</sub>	Gel φ <sub>2</sub>	Gel φ <sub>3</sub>	Sol V <sub>2</sub>	Gel φ <sub>2</sub>	Gel φ <sub>3</sub>	Sol V <sub>2</sub>	Gel φ <sub>2</sub>	Gel φ <sub>3</sub>	Sol V <sub>2</sub>	Gel φ <sub>2</sub>	Gel φ <sub>3</sub>	Sol V <sub>2</sub>	Gel φ <sub>2</sub>	Gel φ <sub>3</sub>
1	.0000	.0000	.1590	.0000	.0000	.1590	.0000	.0000	.1590	.0000	.0000	.1590	.0000	.0000	.1590
	.0442	.0242	.1587	.0429	.0177	.1645	.0457	.0059	.1689	.0455	.0139	.1682	.0455	-	-
	.0889	.0502	.1591	.0864	.0299	.1698	.0913	.0267	.1775	.0915	.0325	.1793	.0910	.0147	.1947
	.1332	.0763	.1607	.1301	.0591	.1763	.1369	.0545	.1887	.1374	.0517	.1914	.1367	.0261	.2191
	.1789	.1022	.1615	.1736	.0863	.1837	.1832	.0745	.1988	.1842	.0690	.2057	.1822	.0375	.2447
.2255	.1357	.1629	.2190	.1083	.1915	.2303	.0957	.2107	.2312	.0887	.2177	.2280	.0472	.2755	
2	.0000	.0000	.1257	.0000	.0000	.1257	.0000	.0000	.1257	.0000	.0000	.1257	.0000	.0000	.1257
	.0439	.0246	.1253	.0427	.0205	.1305	.0454	.0190	.1331	.0453	.0183	.1339	.0451	.0032	.1409
	.0880	.0582	.1264	.0859	.0418	.1372	.0913	.0406	.1410	.0910	.0379	.1437	.0904	.0197	.1602
	.1328	.0837	.1263	.1292	.0709	.1406	.1371	.0629	.1502	.1369	.0624	.1546	.1359	.0334	.1832
	.1768	.1172	.1281	.1732	.0981	.1462	.1833	.0881	.1592	.1833	.0862	.1649	.1817	.0466	.2045
.2235	.1462	.1307	.2175	.1236	.1520	.2301	.1153	.1707	.2295	.1103	.1781	.2275	.0575	.2375	
3	.0000	.0000	.0888	.0000	.0000	.0888	.0000	.0000	.0888	.0000	.0000	.0888	.0000	.0000	.0888
	.0441	.0318	.0883	.0422	.0283	.0921	.0447	.0267	.0936	.0447	.0259	.0958	.0447	.0169	.1026
	.0880	.0625	.0896	.0847	.0562	.0962	.0903	.0544	.1004	.0900	.0501	.1014	.0900	.0344	.1194
	.1328	.0968	.0901	.1291	.0826	.1040	.1344	.0820	.1073	.1353	.0790	.1112	.1349	.0543	.1383
	.1780	.1314	.0910	.1719	.1161	.1045	.1818	.1112	.1158	.1808	.1103	.1208	.1808	.0724	.1592
.2236	.1669	.0916	.2148	.1429	.1073	.2274	.1394	.1246	.2262	.1340	.1277	.2273	.0858	.1806	

$$V_1 = 1 - V_2$$

$$\phi_1 = 1 - \phi_2 - \phi_3$$

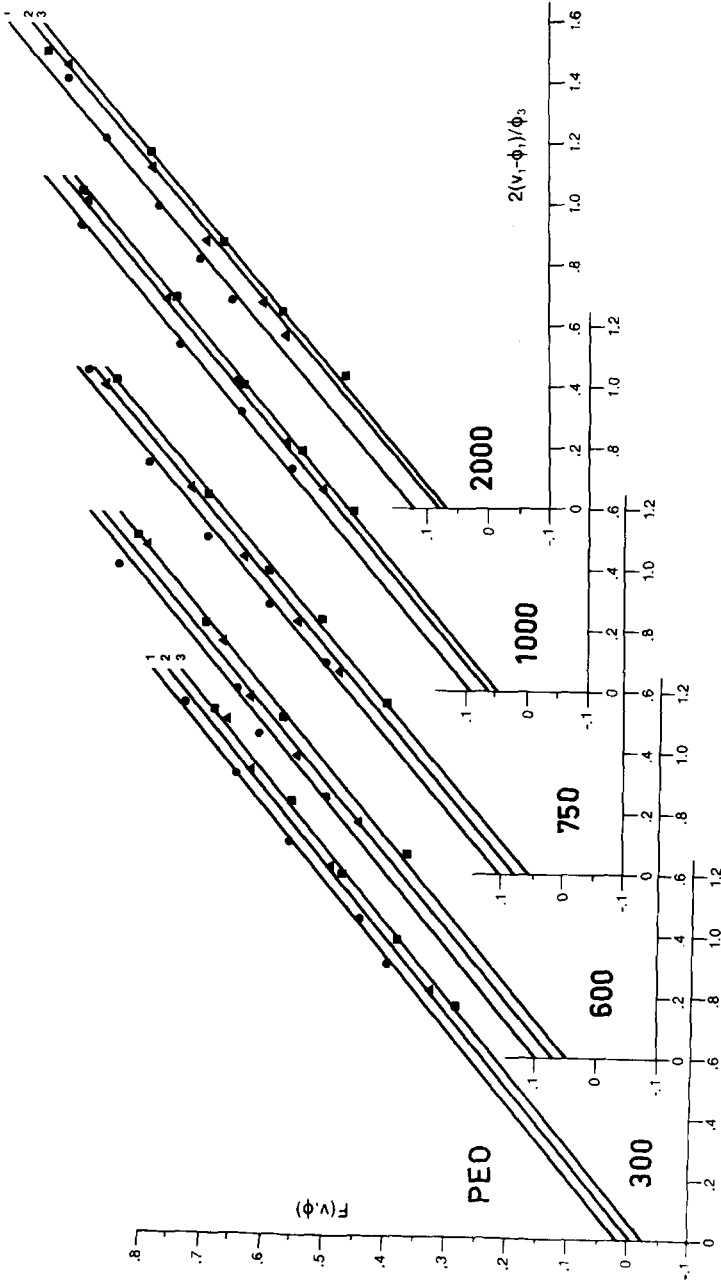


Figure 1: Plots suggested by equation (1) for the gels 1, 2 and 3 where the swelling study is for the system water - PEO and PVP. Starting monomer (I) compositions (mole %): 2.00 for gel 1, 1.00 for gel 2 and 0.50 for gel 3.

the calculations. In Table A we give the equilibrium volume fractions for the PEO linear polymers that penetrated the PVP gels. For  $M_n > 2000$  the gels absorbed no appreciable amount of the linear polymer.

These data are fitted to equation (1) by a least squares method and average  $\chi_{12}$  values for each PEO are tabulated (Avg.  $\chi_{12} = 0.4189 \pm 0.0070$ ): We present the data as a plot of the left side of equation (1) vs.  $2(v_1 - \phi_1)/\phi_3$  as illustrated in Figures 1. The curves are grouped with respect to  $M_n$  of the penetrant, PEO. In these figures all of the lines for each penetrant are drawn with the same slope. These results show, as should be the case, there is no dependence of  $\chi_{12}$  on the cross link density of the gels. Moreover, the slopes for each penetrant used with each gel all fell in a very narrow range of values which fell well within the limits of experimental error. Hence, we have determined the value of  $\chi_{12}$  to be 0.4189 irrespective of  $M_n$  of the PEO. This result is strong evidence that  $\chi_{12}$  is independent of concentration over the concentration range and molecular weight range studied.

PEO	$\bar{V}_{12}$	$\chi_{12}$
300	.0675	.4160
600	.0315	.4143
750	.0263	.4157
1000	.0203	.4282
2000	.0104	.4205

Although we observe the slopes to be independent of penetrant  $M_n$  and gel crosslink density, it is obvious that the intercepts of these curves are dependent on both gel crosslink density and molecular weight of component 2. Although SAKURADA et al.<sup>2</sup> state that the linear relationship should be independent of the  $M_n$  of component 2 it is clearly seen that the intercept from equation (1) is explicitly dependent on  $\bar{V}_{12}$ , the ratio of solvent to linear polymer molar volume. It should be noted that these authors used linear polymers of much higher molecular weight than in the present study. Hence,  $\bar{V}_{12}\chi_{23}$  becomes negligible as compared to  $(\chi_{13} - \chi_{12})$  if  $\chi_{23}$  is not too large. However, in our case  $\bar{V}_{12}\chi_{23}$  is not negligible and must be taken into account. If we call the intercepts from Figures 1 I, we may write,

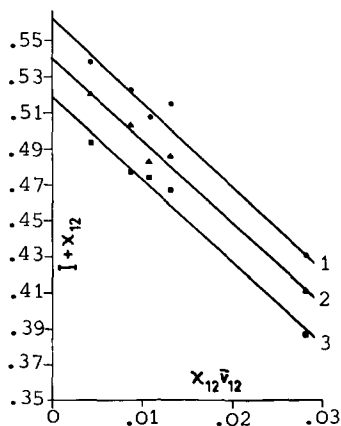
$$I = (\chi_{13} - \chi_{12}) - \bar{V}_{12}\chi_{23}, \quad (2)$$

where  $\chi_{12}$  represents the slopes of these lines. Then, rearranging and solving explicitly for  $(I + \chi_{12})$  we have

$$I + \chi_{12} = \chi_{13} - \left(\frac{\chi_{23}}{\chi_{12}}\right) \chi_{12}\bar{V}_{12}, \quad (3)$$

where we have normalized  $\chi_{23}$  with respect to  $\chi_{12}$  in order to use the values of  $\chi_{12}$  for each PEO molecular weight rather than the average value. In the foregoing table we have presented the  $\chi_{12}$  values for each linear polymer used in the swelling studies and in Figure 2 we have plotted the left side of equation (3) vs.  $\chi_{12}\bar{V}_{12}$  for each hydrogel studied. If  $\chi_{23}$  is independent of crosslink density than all these points should fall on the same straight line having slope of  $-(\chi_{23}/\chi_{12})$  and intercept of  $\chi_{13}$ . If  $\chi_{23}$  is

**Figure 2:** Dependence of the intercept from figs. 1 on  $\bar{V}_{12}$  for PVP gels 1, 2, 3. Slopes yield  $\chi_{23}$  and intercepts give  $\chi_{13}$ .



dependent on crosslink density but independent of molecular weight of the penetrant we should obtain a family of straight lines all having different slopes and intercepts of  $\chi_{13}$  characteristic for the gel crosslink density. It is also conceivable that  $\chi_{23}$  is dependent on both  $M_n$  of component 2 and crosslink density of the gel. It is apparent from Figure 2 that  $\chi_{23}$  is at least dependent on crosslink density as the data points definitely are grouped with respect to gel crosslink density. For penetrant molecular weights  $>300$  we have a fairly good linear correlation. Hence, as may be expected,  $\chi_{23}$  exhibits behavior which supports the idea that the polymer-polymer interaction can be characterized as a structural unit phenomenon which is implied in its definition in the FLORY-HUGGINS theory. Most importantly, we see that  $\chi_{13}$  is quite strongly dependent on crosslink density. This is a crucial point, as it is common practice among workers in the field to assume that the solvent-polymer interaction parameter is independent of crosslink density. This approximation may be valid, but our data indicate it to be valid only for very lightly crosslinked gels. In the table we tabulate the set of interaction parameters found for the three PVP gels and five linear polyethylene glycol polymers used in this study. The  $\chi_{13}$  values reported are extrapolated from the data of Figure 2.

Gel	$\bar{\chi}_{12}$	$\chi_{13}^+$	$\chi_{23}^*$
1	.4189	.5631	1.931
2	.4189	.5412	1.934
3	.4189	.5196	1.913

+ Extrapolated

\* For PEO  $M_n > 600$

### Conclusion

By using the ternary phase relationship for a solvent, crosslinked polymer and linear polymer and extending the treatment first introduced by SAKURADA et al it is possible to determine the FLORY-HUGGINS interaction parameters for the three-component system Water, PEO (linear), and PVP (crosslinked). Moreover, this treatment indicates that the degree of crosslinking in the network polymer is reflected in the value of the interaction parameters associated with the network chains; implying an entropic contribution on the interaction parameter exists.

The interaction parameters for H<sub>2</sub>O-PVP ( $\chi_{13}$ ), H<sub>2</sub>O-PEO ( $\chi_{12}$ ) and PVP-PEO ( $\chi_{23}$ ) are important for us in the development of the thermodynamic equations of state for the two-component gels in which the PEO component is no longer merely a swelling agent but rather a part of the network itself. The correlation between the systems studied here and the two-component networks may well be heavily dependent on our understanding of the interactions elucidated in this paper.

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### References

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